Supporting Material

Hydrogen gas sensing performances of *p*-type Mn₃O₄ nanosystems: the role of built-in Mn₃O₄/Ag and Mn₃O₄/SnO₂ junctions

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S1. Chemico-physical Characterization



Figure S1. Surface silver Auger signal for the Mn₃O₄-Ag specimen.



Figure S2. Deconvolution of surface O1s XP spectra for Mn_3O_4 (a), Mn_3O_4 -Ag (b), and Mn_3O_4 -SnO₂ samples.

S2. Gas Sensing Performances

Response times (τ_{resp} , maximum estimated uncertainty = ±10%) were defined as the times required for the sample resistance to reach 80% of the equilibrium value following hydrogen injection (τ_{resp}) [1-3]. For a hydrogen concentration of 200 ppm at a working temperature of 200°C, typical τ_{resp} values were 360 s, 240 s and 120 s for Mn₃O₄, Mn₃O₄-Ag and Mn₃O₄-SnO₂ specimens, respectively. These results, in line with response data reported in Fig 8 at the same temperature and hydrogen concentration, indicate that functionalization with Ag and SnO₂ produces a faster response with respect to bare Mn₃O₄. The obtained values were lower than the ones previously reported for hydrogen detection by Co₃O₄ [4], NiO [5,6], CuO- [7,8] and MnO₂-based nanosystems [9] under analogous conditions, and comparable to those reported for Pt-Fe₂O₃, Ag/Fe₂O₃ [10] and Ag/ZnO nanomaterials [11]. Nonetheless, the minimization of these values undoubtedly deserves additional research efforts in view of a possible practical use of the developed sensors.

The width of the hole accumulation layer (HAL) in pure *p*-type Mn₃O₄ can be expressed as follows [12-14]:

$$W_{\rm Mn3O4} = \left[\frac{2\varepsilon_{\rm Mn3O4}\Phi}{qN_{\rm Mn3O4}}\right]^{1/2}$$
(S1)

where Φ is the height of the potential barrier established by oxygen adsorption (1.1 eV) [15], ε_{Mn304} is the permittivity of Mn₃O₄ (7.94× ε_0 , where ε_0 is the vacuum dielectric permittivity = 8.854×10⁻¹² C²×N⁻¹×m⁻²) [16], N_{Mn304} is the hole density in Mn₃O₄ (2.25×10²⁴ m⁻³) [17], and q is the electron charge (1.602×10⁻¹⁹ C). Using the above values, the calculated width is W_{Mn304} = 20.6 nm.

Upon functionalization of Mn_3O_4 systems with SnO_2 , the formation of p-n Mn_3O_4/SnO_2 junctions is responsible for the modulation of HAL thickness according to the equation [12,13]:

$$W'_{\rm Mn3O4} = \left[\frac{2\varepsilon_{\rm Mn3O4}\varepsilon_{\rm SnO2}N_{\rm SnO2}V_0}{qN_{\rm Mn3O4}(\varepsilon_{\rm Mn3O4}N_{\rm Mn3O4} + \varepsilon_{\rm SnO2}N_{\rm SnO2})}\right]^{1/2}$$
(S2)

Here, $V_0 = 0.5 \text{ eV}$ is the contact potential difference between SnO₂ and Mn₃O₄, calculated as the difference between the single oxide work function (WF) values [18,19], whereas $\varepsilon_{\text{SnO2}}$ (18.2× ε_0) and N_{SnO2} (3.6×10²⁴ m⁻³) are the permittivity and the electron concentration in SnO₂, respectively [20]. The calculation yields $W'_{\text{Mn3O4}} = 12.4 \text{ nm}.$

In the case of Mn₃O₄-Ag systems, the occurrence of a finite metal/semiconductor junction [21], as well as the partial Ag oxidation (as demonstrated by XPS analyses, see the main paper text), prevent from a detailed and straightforward numerical calculation.

Sensor	H ₂ concentration (ppm)	Temperature (°C)	$\frac{\text{Response}}{\left(\frac{R_G - R_A}{R_A}\right) \times 100}$	Ref.
Mn3O4-Ag	200	300	15	Present study
Mn3O4-SnO2	200	200	19	Present study
CuO	200	300	10	[22]
NiO	1000	150	≈0	[23]
BiFeO ₃	500	300	5	[24]
C03O4	200	600	≈0	[25]
NixCo3-xO4	200	600	1	[25]
MnO2-rGO	500	85	0.4	[26]
MnO2-MWCNTs	3×10 ⁴	220	7.5	[27]
MnO ₂ -WO ₃	50-200	200	0	[28]

Table S1. Comparison of hydrogen sensing properties of the present Mn₃O₄-based sensors with selected representative literature works. rGO = reduced graphene oxide; MWCNTs = multi-walled carbon nanotubes.



Figure S3. Gas responses as a function of H₂ concentration for bare and functionalized Mn₃O₄ sensors. Working temperature = 200°C for Mn₃O₄-SnO₂; 300°C for Mn₃O₄ and Mn₃O₄-Ag.



Figure S4. Gas responses to fixed CO₂, CH₄, and H₂ concentrations (500 ppm, 250 ppm and 200 ppm, respectively) for Mn₃O₄-Ag and Mn₃O₄-SnO₂ sensors.

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